



Interface engineering of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ heterostructure nanosheets as a highly active electrocatalyst for overall water splitting and Zn- H_2O cell

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ABSTRACT

The key in developing a low-cost, high-efficiency electrocatalyst for hydrogen generation is not only cutting the cost by avoiding noble metals but also utilizing the structure–function relationship to expose the maximum amounts of active sites on the surface by increasing the interface between the active components. Here, we demonstrated full-cycle synthesis, characterization, and optimization of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ electrocatalyst on carbon-supported by density functional theory (DFT) calculations. The DFT calculation revealed a significant charge accumulation at the interface between $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 , suggesting the possibility of a strong synergy. As expected, electrochemical studies have shown a bifunctional $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst with low overpotential and durability towards hydrogen/oxygen evolution reactions (HER/OER) in alkaline electrolytes and robust overall water splitting performance at high current densities. In addition, the optimized $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst is also used in a Zn- H_2O cell and displayed a power density exceeding Pt/C with the long-term stability of up to 170 h. The excellent electrochemical performance is the outcome of the better charge mobility at the interface resulted in the unique synergy between the active components.

1. Introduction

Electrochemical water splitting is considered to be the cleanest and sustainable method for high-quality hydrogen production that heavily depends on catalysts to facilitate H_2 generation and subsequent utilization [1–3]. The key challenges in catalyst development are to overcome the slow reaction kinetics resulted from high overpotential [4–8]. Moreover, large-scale H_2 production to grid-level requires catalysts that are easy to prepare, cheap, and comparable to the state-of-the-art noble-metal-based catalysts [9–13]. Therefore, the development of low-cost, high-efficiency transition metal-based hydrogen fuel catalysts is of great significance [14].

In the past few decades, significant progress has been made in utilizing transition metal alloys and metallic compounds such as sulfides [15,16], phosphides [17,18], nitride compounds [19–22], borides [23,24], oxides [25] etc., as potential electrocatalysts for water splitting. Among all of these materials, transition metal nitrides and oxides exhibit great potential due to their excellent electrical conductivity and

corrosion resistance [26–28]. Although few heterogeneous interfaces about transition metal nitrides and oxides have been reported, the designed catalysts still possess relatively high overpotentials that hinder their further application. As an emerging new strategy, interface engineering creates interfacial synergy through optimizing the electronic structure and thereby improves the catalytic activity of water splitting [11,18,29]. Hu et al. reported a heterostructure catalyst of $\text{Ni}_3\text{N}/\text{Ni}_{0.2}\text{Mo}_{0.8}\text{N}$ microspheres that exhibited high HER and OER performances by establishing abundant interfaces between two components [30]. Afterwards, Cao et al. demonstrated a multifunctional Mo-doped $\text{NiCo}_2\text{O}_4/\text{Co}_{5.47}\text{N}$ heterostructure electrocatalyst that presented remarkable catalytic activity toward the OER ($\eta_{50} = 310$ mV) and HER ($\eta_{50} = 170$ mV), respectively [31]. Obviously, bifunctionality permits the cost reduction plus intrinsic stability that extends the catalysts lifetime through in situ electrochemical regeneration by bias-switching [32,33]. Furthermore, the bifunctionality also hints at the structure–function further guiding the process parameters toward the optimal points [34]. Nevertheless, there are still rooms to improve the catalyst

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performance and stability especially at high current densities ($>100 \text{ mA cm}^{-2}$) [35–37]. Regulating the morphology of catalyst to maximize the active site utilization is feasible through a rational design and unique preparation method, especially stereo-assembly of the heterostructure. Additionally, contrary to the electrochemical water splitting that requires additional energy to generate H_2 , the Zn- H_2O fuel cell is capable of simultaneous hydrogen production and electricity generation using the energy generated from Zn oxidation [38,39]. However, transition metal-based nitrides have not yet been explored in Zn- H_2O fuel cells.

In this work, the hetero-structured $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ nanoparticles electrocatalysts were *in-situ* assembled on CC to forming nanosheet through a simple ion-exchange hydrothermal process followed by nitrating treatment. The microstructure and chemical state of the catalysts were characterized and analyzed. XRD and TEM proved the existence of $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 crystals and the formation of heterogeneous interfaces. The electrochemical performance of the catalyst in an alkaline medium was further studied. The experimental results show that the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst requires overpotentials of only 63 mV (HER) and 283 mV (OER) to drive the current density of 10 mA cm^{-2} , respectively. Furthermore, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst used as a cathode in alkaline Zn- H_2O fuel cell and displayed a power density of up to 20 mW cm^{-2} and superior to that of commercial Pt/C (10.5 mW cm^{-2}) under the same condition. More importantly, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ reveals robust stability of keeping 170 h for H_2 generation. Theoretical calculations demonstrate that the synergy between $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 components at the interface can reduce the energy barrier of reaction intermediates, and facilitate the release of H_2 and O_2 , thereby improving the catalytic activity. This study is focused on the overall performance and the cost-effectiveness of bifunctional catalyst. Moreover, the scalability and stability at high current density for actual industrial application. The studies on the heterojunction of cobalt-molybdenum metal nitrides and oxides have been rare, especially as a dual-function electrocatalyst for the HER and OER. Finally, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ synthesized by the ion exchange hydrothermal method has a nanosheet structure composed of nanoparticles that creates a sufficient heterogeneous interface, thereby better performance.

2. Experimental details

2.1. Materials

Cobalt nitrate hexahydrate ($\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, 99.0%), sodium molybdate dehydrate ($\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$, 99.0%), urea ($\text{CH}_4\text{N}_2\text{O}$, $\geq 99.0\%$) were analytical reagents and used without further purification. Commercial Pt/C (20 wt% for platinum) was purchased from Alfa Aesar.

2.2. Synthesis of CoCH/CC nanosheets

Typically, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (2 mmol) and urea (10 mmol) were dissolved in 35 mL water under continuously stirring for 30 min. The solution was transferred to a 50 mL autoclave lined with polytetrafluoroethylene, and the cleaned carbon cloth (CC) was placed in the autoclave at an angle. The autoclave was then sealed and heated at 120°C for 6 h. After cooling, the sample of $\text{Co}(\text{CO}_3)_{0.5}(\text{OH})_{0.11} \cdot \text{H}_2\text{O}/\text{CC}$ (CoCH/CC) was taken out and thoroughly washed with water and ethanol and then dried at 60°C (loading: 1.4 mg cm^{-2}).

2.3. Synthesis of CoCH/CoMoO₄ nanosheets on CC

By changing the amount of $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$ (100, 150, and 200 mg mL^{-1}), CoCH/CoMoO₄ samples with different Mo content were prepared. Using a similar hydrothermal method, $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$ was dissolved in 35 mL of water. After continuous stirring for 30 min, the solution was transferred to a 50 mL autoclave lined with polytetrafluoroethylene. The precursor CoCH/CC was placed in the autoclave at an angle and heated at 130°C for 6 h. Similarly, after cooling, the CoCH/

CoMoO₄ was taken out and washed with water and ethanol thoroughly, and dried overnight (loading: 2.3 mg cm^{-2}).

2.4. Synthesis of Co₂N_{0.67}/CoMoO₄ nanosheets on CC (Co₂N_{0.67}/CoMoO₄)

The obtained CoCH/CoMoO₄ was placed in a tube furnace and annealed at 400°C for 5 h with a ramp rate of 5°C min^{-1} under a flow of ammonia gas, respectively. The obtained sample was denoted as $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ (loading: 2.1 mg cm^{-2}).

2.5. Synthesis of Co₂N_{0.67} nanosheets on CC (Co₂N_{0.67}/CC)

The obtained CoCH was placed in a tube furnace and annealed at 400°C for 5 h with a ramp rate of 5°C min^{-1} under a flow of ammonia gas. The obtained sample was denoted as $\text{Co}_2\text{N}_{0.67}$.

2.6. Synthesis of CoMoO₄ nanosheets on CC (CoMoO₄/CC)

The synthesis process of CoMoO₄/CC is similar to that of CoCH/CC, except that urea (10 mmol) is replaced by $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$ (2 mmol). The obtained product was placed in a tube furnace and annealed at 350°C for 2 h with a ramp rate of 5°C min^{-1} under a flow of nitrogen gas.

2.7. Synthesis of RuO₂ and 20 wt% Pt/C electrodes

The $\text{RuCl}_3 \cdot 3\text{H}_2\text{O}$ was directly calcined at 400°C for 3 h in the air to obtain RuO₂ powders. And then 2 mg RuO₂ or 20 wt% commercial Pt/C was dispersed into a mixture of 100 μL deionized water, 100 μL ethanol and 5 μL 5 wt% Nafion, respectively. The mixture was ultrasonically treated for at least 30 min to form a uniform catalyst ink, then dropped onto the surface of CC (1 cm \times 1 cm) and dried naturally in the air. The corresponding XRD pattern are indexed to RuO₂ (JCPDS: 40–1290), confirming the successful preparation of RuO₂. The SEM image indicates that RuO₂ exists in the form of particles (Fig. S2).

2.8. Electrochemical measurements

The electrocatalytic activity was performed on a Biologic VMP3 electrochemical work station with a standard three-electrode system in alkaline. The catalysts, graphite plate, and calomel electrode were used as working, counter and reference electrodes, respectively. The cyclic voltammetry (CV) test was performed with a sweep rate of 5 mV s^{-1} at a potential of -0.9 to -1.5 V and 0 to 0.8 V for HER and OER, respectively. Linear sweep voltammetry (LSV) was carried out at a scan rate of 1 mV s^{-1} , and the electrochemical impedance spectroscopy (EIS) measurements were performed near the onset potential in the frequency range from 200 kHz to 10 mHz. The potential of E_{RHE} was calibrated by the equation of $E_{\text{RHE}} = E_{\text{SCE}} + 0.241 + 0.059 \text{ pH}$, consistent with the calibration results of the standard hydrogen electrode (RHE) (Fig. S1). All the LSV polarization curves were corrected against iR compensation. The electrochemical double-layer capacitance (C_{dl}) measured by cyclic voltammetry at different scan rates (1, 2, 4, 6, 8, 10 mV s^{-1}) to estimate the electrochemically active surface area (ECSA). The overall water splitting test was performed by a two-electrode system in the potential range of 0 ~ 2.0 V with a scan rate of 5 mV s^{-1} in 1.0 M KOH solution.

2.9. Zn-H₂O cell measurements

The Zn- H_2O cell was assembled according to the following process: typically, the cathode electrode was a CC (1 \times 1 cm^2) loaded with $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst (2.1 mg). Then a polished zinc foil was used as the anode, 6 mol L^{-1} KOH and 0.2 mol L^{-1} Zn(Ac)₂ were used as an electrolyte, and the electrolyte was saturated with N₂ before the test for suppressing the oxygen reduction reaction. For comparison, 2.1 mg Pt/C

catalyst ink was prepared by loading on a $1 \times 1 \text{ cm}^2$ CC (the catalyst was mixed with 5% Nafion solution and water/isopropanol solution (1:3 (v/v)) to prepare catalyst ink). All electrochemical tests were conducted on electrochemical workstation (CHI 760E) under ambient conditions. The stability was tested by using the LAND battery testing system (BT2016A) at a current density of 5 mA cm^{-1} with 20 min per cycle (10 min charge and 10 min discharge). The specific capacity was calculated from the following equation:

$$\text{specific capacity} = \frac{\text{discharge current} \times \text{time}}{\text{weight of consumed Zinc}}$$

3. Results and discussion

3.1. Synthesis strategy and structural analysis

The fabrication processes of hierarchical $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ nanosheets are a three-step route, as shown in Fig. 1a. The Co (CO_3) $_{0.5}(\text{OH})_{0.11} \cdot \text{H}_2\text{O}$ (CoCH) nanosheets precursor on the CC was first prepared by a hydrothermal method followed by a hydrothermal ion-exchange process to make CoCH/CoMoO₄ nanosheets, which were

further processed under NH_3 atmosphere at 400°C to generate the final composite of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. The corresponding SEM image shows uniform distribution of nanosheet arrays on the CC surface after the hydrothermal process (Fig. S3a), and the X-ray diffraction (XRD) patterns is indexed to CoCH (JCPDS: 48–0083) (Fig. 1b) [18], which showed a stable morphology after the consecutive hydrothermal treatment (Fig. S3b). However, an additional CoMoO_4 crystal phase (JCPDS: 25–1434) (Fig. S4b) appeared, indicating the formation of CoMoO_4 species through the ion exchange process.

As shown in Fig. 1b, the XRD pattern indicated that the CoCH characteristics diffraction peaks disappeared completely after ammonia treatment at 400°C . In contrast, the characteristic diffraction peak of $\text{Co}_2\text{N}_{0.67}$ (JCPDS: 06–0691) was observed, confirming that CoCH has been converted to $\text{Co}_2\text{N}_{0.67}$ [40]. Both the CoCH and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ samples show a prominent carbon peak between 20 and 30° . The characteristic diffraction peak of molybdenum nitride isn't detected since molybdenum nitride formation requires a much higher temperature ($\sim 700^\circ\text{C}$), and the catalysts were prepared at 400°C in this study corresponding obvious peaks at $14.2, 28.5, 43.4, 59.6, 61.2,$ and 62.8° can index well with CoMoO_4 (JCPDS: 25–1434) [41]. The control

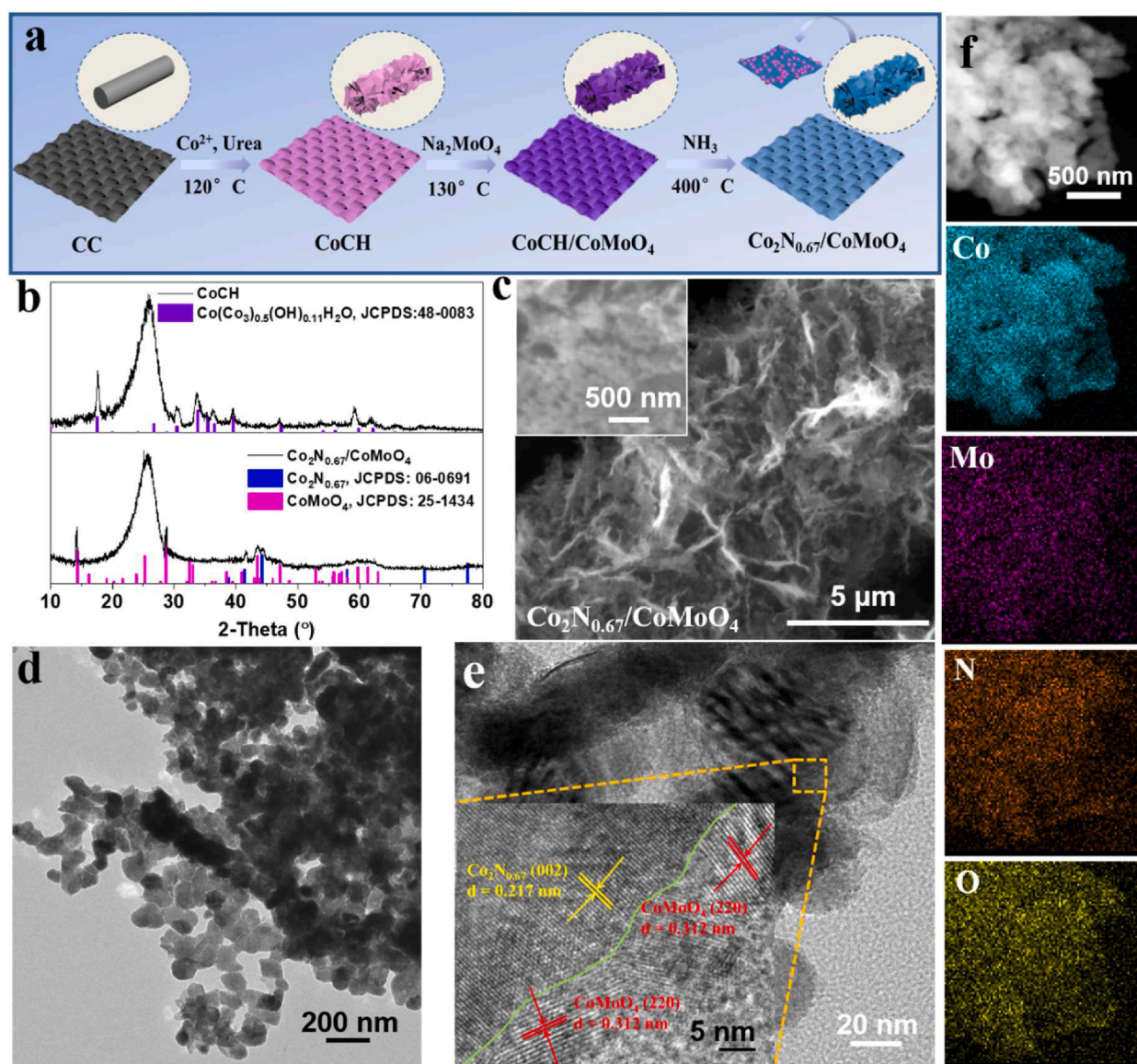


Fig. 1. (a) Schematic illustration of the synthesis of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (b) XRD patterns of CoCH and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (c) SEM images of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (d) TEM image, (e) High-resolution TEM image, and (f) HAADF-STEM image and corresponding elemental mappings of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ (color online).

experiment (nitriding the CoCH precursor and hydrothermal of CoNO_3 and NaMoO_4) also displayed the formation of the $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 crystal phase, respectively (Fig. S4a and S4c). The SEM image indicated that $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ maintained the nanosheet structure after annealing, except that the surface of the nanosheet became rough in virtue of ammonia etching (Fig. 1c). The N_2 adsorption/desorption isotherm of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ displays a typical type III behavior and the Brunauer-Emmett-Teller (BET) surface area of $27.5 \text{ m}^2 \text{ g}^{-1}$. Barrett-Joyner-Halenda (BJH) pore size distribution curve (Fig. S4d) reveals that $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ has a mesoporous structure with a pore size of about 3.8 nm.

The high-resolution SEM and transmission electron microscopy (TEM) images displayed that the nanosheet of the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ is composed of numerous nanoparticles with average particles size of 100 nm at high resolution (Fig. 1d) [42,43]. As shown in the high-resolution TEM image of Fig. 1e, one of the noticeable lattice spacings of 0.217 nm corresponds to the (002) crystal plane of $\text{Co}_2\text{N}_{0.67}$, and the $d = 0.312 \text{ nm}$ is corresponding to the (220) crystal plane of CoMoO_4 . Interestingly, the apparent grain boundaries at high magnification were not reflected on lower resolution elemental mapping, consistent with other reports [2,44,45]. Moreover, there is an observable interface between the $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 crystal phases, indicating the existence of heterostructures. The corresponding element mapping images of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ (Fig. 1f) reveal the uniform distribution of Co, Mo, N, and O throughout the catalyst.

3.2. Chemical state analysis

X-ray photoelectron spectroscopy (XPS) was used to study the surface electronic valence and chemical composition. As shown in Fig. S5a, the XPS spectrum shows that the surface of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ is mainly composed of Co, Mo, N, and O elements. The high-resolution C 1s spectrum is deconvoluted into four peaks at 284.0 (C=C), 284.8 (C-C) and 285.3 eV (C-N) and 286.1 eV (C-O) (Fig. S5b) [46,47]. The Co (II) and Mo (VI) peaks are observed in the oxide samples (Fig. 2a, b). The peaks at 779.4 and 781.3 eV in Co $2p_{2/3}$ region can be attributed to Co-N

and Co(II), respectively. (Fig. 2a) [26,40]. There is a strong Co-N peak that appeared in $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ after NH_3 treatment indicates N doping. The high-resolution XPS spectrum of Mo 3d of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ and $\text{CoCH}/\text{CoMoO}_4$ are shown in Fig. 2b. The peak at 232.6 eV is assigned to Mo (VI) of CoMoO_4 [46,48]. Obviously, the content of Co^{2+} is increasing with the introduction of Mo, confirming that $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ possesses more active sites [49]. The N 1s spectrums are shown in Fig. 2c. The peaks at 397.5 and 398.7 eV are assigned to the Co-N and pyridinic-N of the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ sample [4,50]. Meanwhile, the O 1s spectrum is deconvoluted into three peaks at 530.2, 531.8, and 533.0 eV assigned to the Metal-O, O-vacancies and adsorbed O species from $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$, respectively (Fig. 2d) [51,52].

3.3. Electrocatalytic HER analysis

The HER behavior of the catalysts with an area of $1 \times 1 \text{ cm}^2$ was measured using a standard three-electrode system in 1.0 M KOH at a scan rate of 2 mV s^{-1} (Fig. S6). As shown in Fig. 3a-b, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst showed an HER overpotential of $63 \text{ mV}@10 \text{ mA cm}^{-2}$, which is comparable to that of the commercial Pt/C. Whereas, it only requires 315 mV to reach 1000 mA cm^{-2} , indicating a high potential for industrial applications. As expected, $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ displayed better electrocatalytic activity that surpassed those of CoMoO_4 , CoCH , $\text{Co}_2\text{N}_{0.67}$, $\text{CoCH}/\text{CoMoO}_4$ catalysts, and other similar transition metal-based catalysts (Table S2). Generally, the HER kinetic and the rate-determining steps are investigated by Tafel plots. The Tafel slope of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ was 57.5 mV dec^{-1} , lower than those of CoCH ($153.5 \text{ mV dec}^{-1}$), CoMoO_4 ($175.2 \text{ mV dec}^{-1}$), $\text{Co}_2\text{N}_{0.67}$ ($127.8 \text{ mV dec}^{-1}$), and $\text{CoCH}/\text{CoMoO}_4$ ($146.7 \text{ mV dec}^{-1}$), revealing a Volmer-Heyrovsky reaction pathway is the rate-determining step for $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ (Fig. 3c) [18,46]. Further, EIS results demonstrated that the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ displayed the lowest charge-transfer resistance (R_{ct}) among all samples except Pt/C, implying a fast charge transfer mechanism (Fig. S7). Particularly, XRD patterns confirmed that as prepared samples different Na_2MoO_4 concentration have similar composite structure of $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 (Fig. S8). The LSV curve shows that $\text{Co}_2\text{N}_{0.67}/$

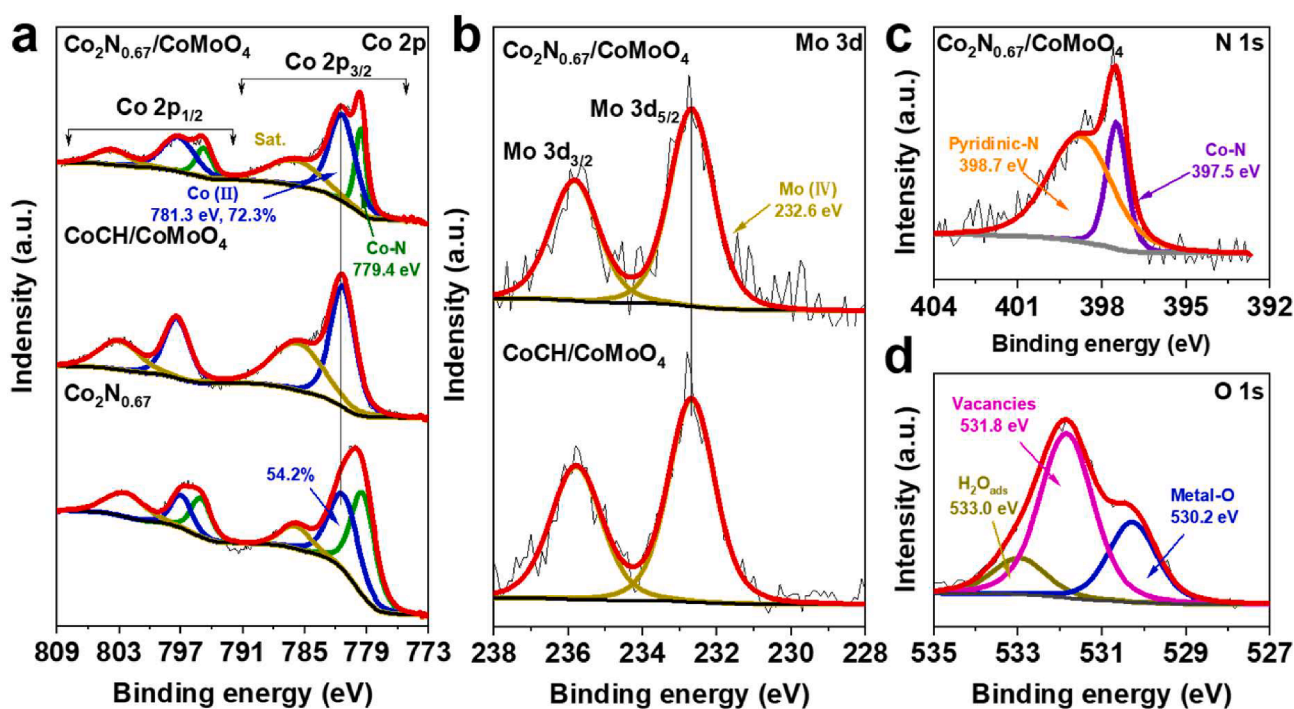


Fig. 2. High-resolution XPS spectra of (a) Co 2p from $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$, $\text{CoCH}/\text{CoMoO}_4$ and $\text{Co}_2\text{N}_{0.67}$. (b) Mo 3d regions from $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ and $\text{CoCH}/\text{CoMoO}_4$, respectively. (c) N 1s and (d) O 1s regions from $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$.

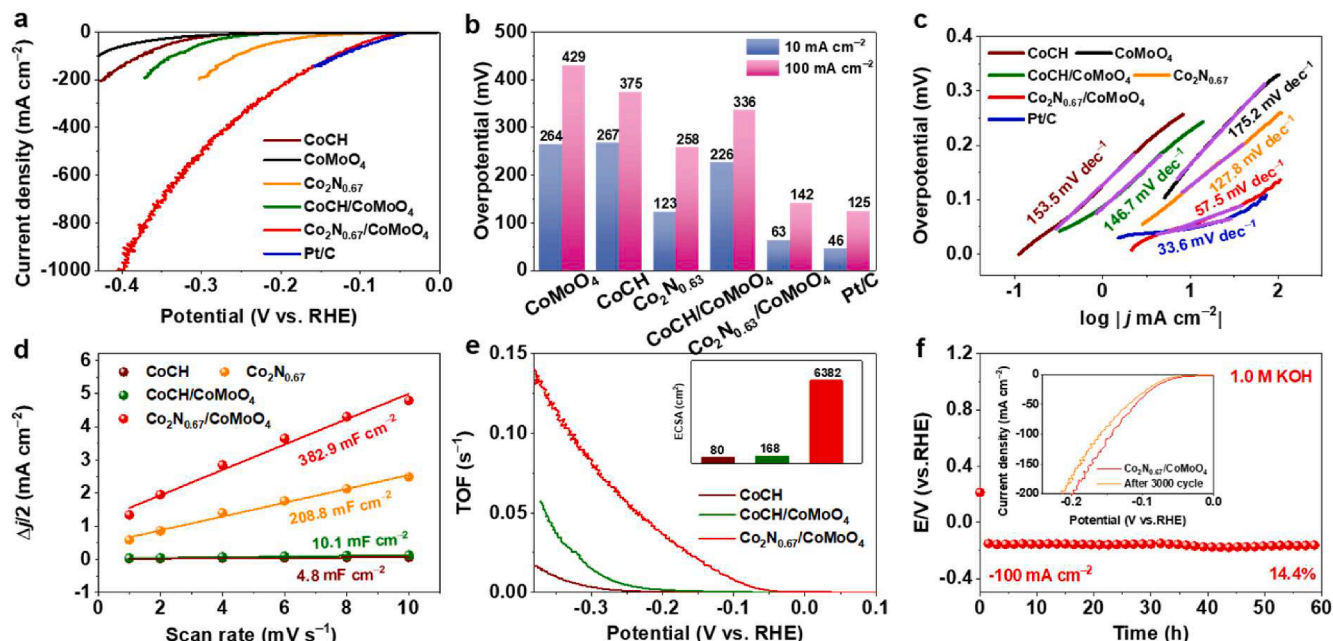


Fig. 3. Electrocatalytic HER performance. (a) LSV polarization curves, (b) the comparison of overpotential at 10 and 100 mA cm⁻² for Co₂N_{0.67}/CoMoO₄ and other references, (c) corresponding Tafel slopes, (d) summarized double-layer capacitance (C_{dl}) of different catalysts in 1.0 M KOH. (e) Potential-dependent TOF curves (Inset: a bar chart of the ECSA) of CoCH, CoCH/CoMoO₄, and Co₂N_{0.67}/CoMoO₄, respectively. (f) Durability tests of Co₂N_{0.67}/CoMoO₄ for HER at -100 mA cm⁻² and (Inset: Polarization curve before and after 3000 cycles of cyclic voltammetric stability test in 1.0 M KOH solution).

CoMoO₄ has the best HER activity when the adding amount of Na₂MoO₄ is 150 mg mL⁻¹ (Fig. S9). The study indicates that the HER performance is related to the available interface between Co₂N_{0.67} and CoMoO₄. The best HER performance was achieved when the NaMoO₄ used is 0.15 g, and the ratio of the two components is close to 1:1, which may lead to the most accessible interface (Table S1) [18]. To further explore the influencing factors of catalytic kinetics, the electrochemical double-layer capacitance (C_{dl}) was measured by cyclic voltammetry (CV) measurements in the non-Faradaic region at different scan rates (Fig. S10). In Fig. 3d, the C_{dl} of Co₂N_{0.67}/CoMoO₄ is 382.9 mF cm⁻², larger than those of CoCH (4.8 mF cm⁻²), Co₂N_{0.67} (208.8 mF cm⁻²), and CoCH/CoMoO₄ (10.1 mF cm⁻²). Generally, electrochemically active surface area (ECSA) is directly proportional to C_{dl} , and is used to estimate the number of active sites. To be specific, the Co₂N_{0.67}/CoMoO₄ retains the maximum ECSA of 6382 cm², which surpasses CoCH (80 cm²), CoCH/CoMoO₄ (168 cm²), Co₂N_{0.67}/CoMoO₄-100 (3553 cm²), and Co₂N_{0.67}/CoMoO₄-200 (5106 cm²), suggesting that the Co₂N_{0.67}/CoMoO₄ catalyst has more active sites (inset: Fig. 3e and Fig. S11). The more active sites may be caused by the generation of oxygen vacancies and the formation of the abundant interface on rough nanosheets after nitriding treatment, which can increase the contact area between the catalyst and the electrolyte to provide more active sites for HER point [53]. Notably, the LSV polarization curves are normalized by ECSA indicating that Co₂N_{0.67}/CoMoO₄ has the lowest overpotential (Fig. S12) [54,55]. This result confirms that the strong interfacial interaction between Co₂N_{0.67} and CoMoO₄ species increases ECSA and enhances intrinsic activity. The turn over frequency (TOF) value of Co₂N_{0.67}/CoMoO₄ greatly increased compared with CoCH and CoCH/CoMoO₄, similarly meaning that the Co₂N_{0.67}/CoMoO₄ catalyst increased intrinsic activity after nitriding treatment, thus improving the catalytic activity (Fig. 3e) [53]. Considering the high performance of Co₂N_{0.67}/CoMoO₄, we carried out a long-term stability test at a constant current density of -100 mA cm⁻², and the results showed that the current density has declined by approximately 14.4% after 60 h (Fig. 3f). In addition, the stability of the catalyst was further explored through multi-cycle CV testing (inset: Fig. 3f) and the polarization curve dropped slightly after 3000 cycles. The degradation of catalytic performance can

be attributed to part of the morphology agglomeration (Fig. S13), slight changes in crystal structure (Fig. S14) and chemical valence (Fig. S15), as well as the combined effect of the dissolution of CoMoO₄ [56,57].

3.4. Electrocatalytic OER analysis

The OER catalytic performance was also evaluated by linear sweep voltammetry (LSV) polarization in 1.0 M KOH. As shown in Fig. 4a, the overpotential required for Co₂N_{0.67}/CoMoO₄ catalyst is 283 mV at a current density of 10 mA cm⁻², which is slightly higher than that of commercial RuO₂, but far less than those of other comparative OER catalysts (Table S3). Remarkably, the performance of Co₂N_{0.67}/CoMoO₄ is superior to RuO₂ at a high current density (>40 mA cm⁻²) due to a mesoporous nanosheet structure that has better conductivity and small bubble size [58,59]. Tafel slopes are shown in Fig. 4b, the Co₂N_{0.67}/CoMoO₄ shows a smaller Tafel slope among all other OER catalysts, suggesting faster reaction kinetics in the OER process, which can be attributed to the synergistic effect between Co₂N_{0.67} and CoMoO₄ [4,60]. Besides, the Co₂N_{0.67}/CoMoO₄ has a smaller R_{ct} value than CoCH, CoMoO₄, Co₂N_{0.67}, and CoCH/CoMoO₄, revealing a faster charge transfer than others (Fig. 4c). Notably, the microscopic morphology changes little after stability (Fig. S16). The XRD and XPS analysis indicated that the chemical valences of the elements have not changed evidently after the stability test (Fig. S15 and Fig. S17), and Energy dispersive spectroscopy (EDS) shows that only 0.1% of Mo species remained on the surface (Fig. S18), which may be caused by the dissolution of Mo species during the OER process [61].

3.5. DFT calculations

Using density functional theory (DFT) calculations to further understand the excellent HER and OER activity. The differential charge density of the heterojunction shows that there is obvious charge accumulation at the interface of the two materials, indicating that Co₂N_{0.67} and CoMoO₄ have strong interaction and synergistic effects (Fig. 5a). To further elucidate the interaction between Co₂N_{0.67} and CoMoO₄ in heterojunction catalysts, the density of states (DOS) of Co₂N_{0.67},

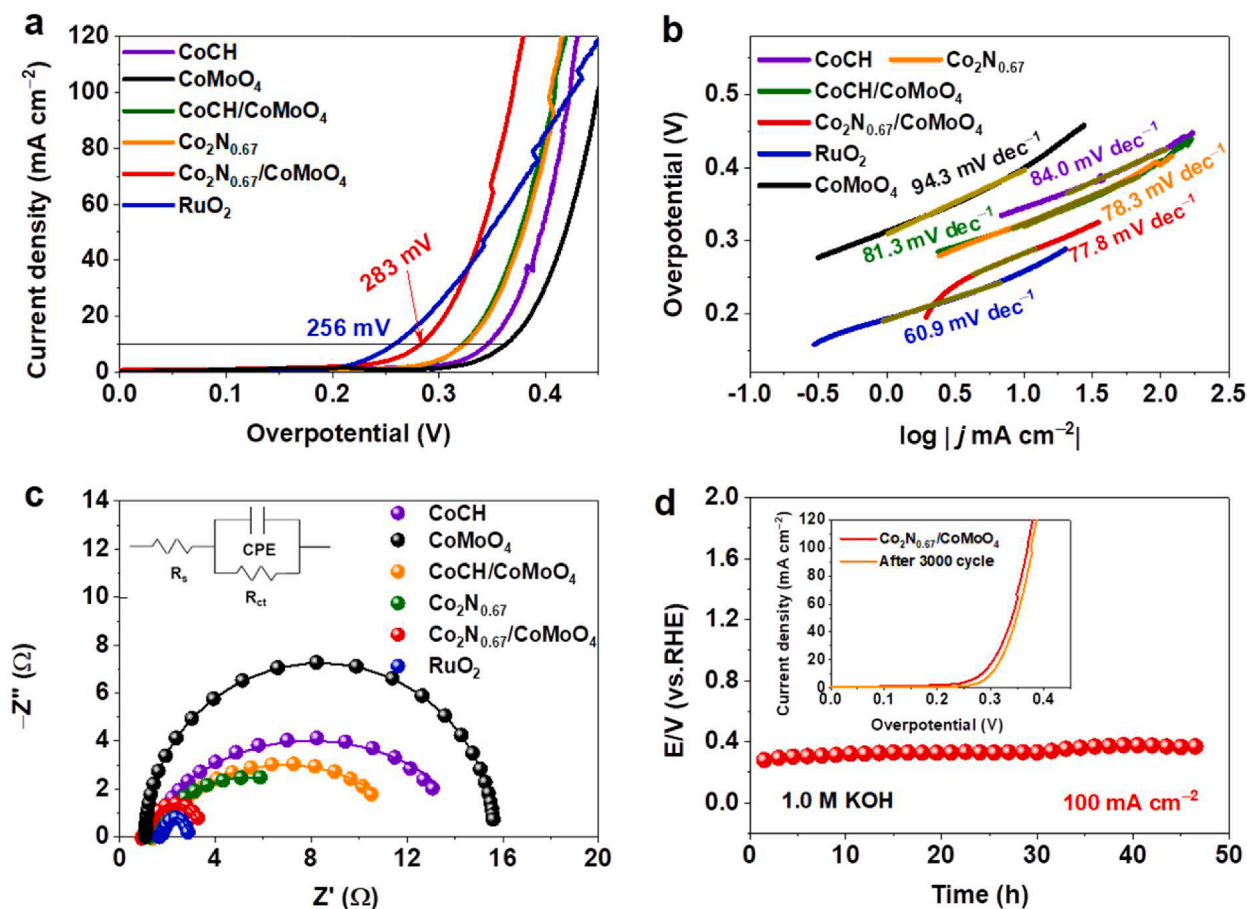


Fig. 4. Electrochemical OER performance in 1.0 M KOH. (a) LSV polarization curves, and (b) the corresponding Tafel slopes of different catalysts. (c) Electrochemical impedance spectroscopy (EIS) of different catalysts. (d) Durability tests of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ for HER at 100 mA cm^{-2} and (Inset: Polarization curve before and after 3000 cycles of cyclic voltammetric stability test in 1.0 M KOH solution).

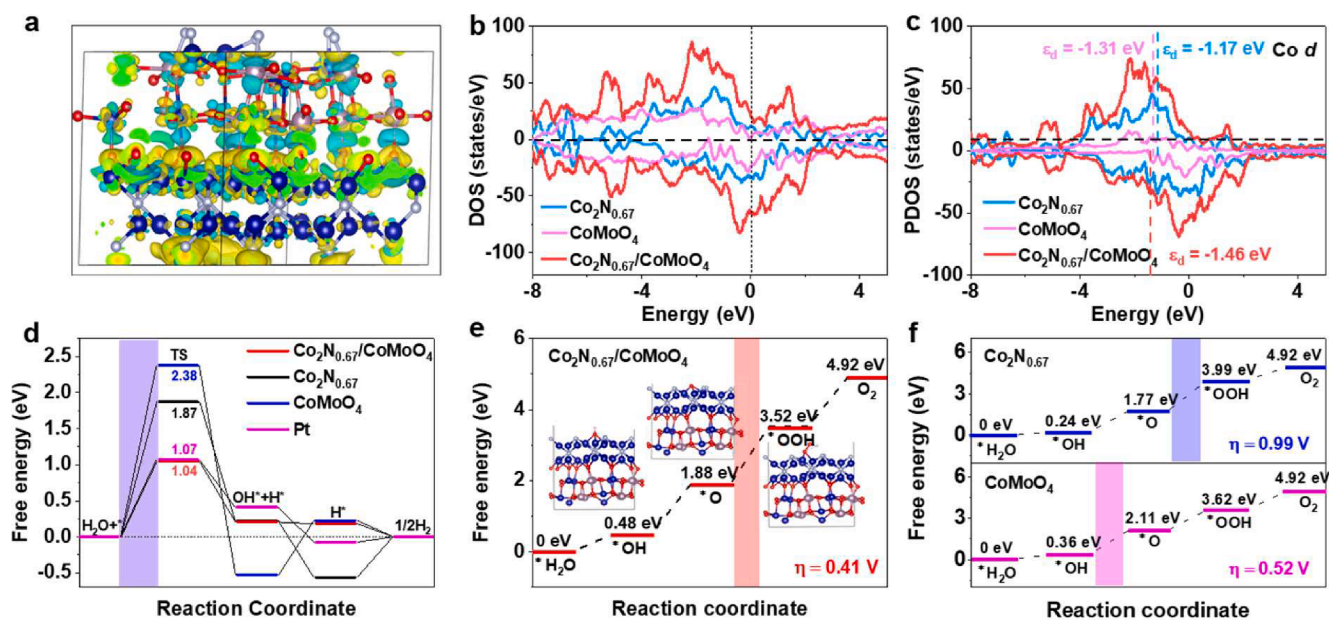


Fig. 5. Density functional theory (DFT) calculations on $\text{Co}_2\text{N}_{0.67}$, CoMoO_4 and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (a) Schematic diagram of differential charge structure model. The blue, pink, red, and silver balls represent the Co, Mo, O, and N atoms, respectively. (b) The density of states (DOS) of $\text{Co}_2\text{N}_{0.67}$, CoMoO_4 , and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (c) Projection DOS (PDOS) of Co 3d track of $\text{Co}_2\text{N}_{0.67}$, CoMoO_4 , and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. The dotted line represents the center of the d-band of the catalyst. (d) Free energy diagram of H adsorption (ΔG_{H^+}) for $\text{Co}_2\text{N}_{0.67}$, CoMoO_4 , and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$. (e, f) Free-energy diagrams of $\text{Co}_2\text{N}_{0.67}$, CoMoO_4 , and $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ for OER.

CoMoO₄, and Co₂N_{0.67}/CoMoO₄ structures were calculated. As shown from Fig. S19, Co atom in Co₂N_{0.67} shows a higher electron density near the E_F and more positive E_d, which can infer this Co species act as main active sites [62]. The Co atom in CoMoO₄ serves as a backup electron reservoir to transfer the electron of Mo species and protect the valence of the Co of Co₂N_{0.67} [63]. It is note that the *d-d* coupling between two Co species enabled the higher electron density of Co₂N_{0.67}/CoMoO₄ near the Fermi level, which can accelerate the electron transfer on the surface to improve the performances of water splitting (Fig. 5b) [64]. In addition, the *d*-band center position of the Co atom in the Co₂N_{0.67}/CoMoO₄ structure is moved down to -1.46 eV compared with the single Co₂N_{0.67} (ε_d = -1.17 eV) and CoMoO₄ (ε_d = -1.31 eV) which led to more optimal H_{ads} and thereby pronouncedly be conducive to following reactions (Fig. 5c) [65].

The intrinsic catalytic activity during the alkaline HER process was further investigated by Gibbs free energy of Pt, Co₂N_{0.67}, CoMoO₄, and Co₂N_{0.67}/CoMoO₄ (Fig. 5d). Generally, the activation of the H₂O molecule for the transition state (TS) is the rate-determining step for alkaline HER [66,67]. Obviously, Co₂N_{0.67}/CoMoO₄ heterostructure demonstrates a desired activation energy barrier of 1.04 eV lower than Pt (1.07 eV), Co₂N_{0.67} (1.87 eV), and CoMoO₄ (2.38 eV), implying the combination of Co₂N_{0.67} and CoMoO₄ can remarkably accelerate the disintegration of H₂O to achieve a faster HER kinetics. A desirable HER catalyst should demonstrate a moderate hydrogen binding energy (HBE) in the Heyrovsky step to adjust the adsorption of H* and the following desorption of H₂ [68,69]. The Co₂N_{0.67}/CoMoO₄ heterostructure shows a more suitable HBE value of 0.18 eV comparing the Co₂N_{0.67} and

CoMoO₄ of -0.56 and 0.23 eV, implying the Co₂N_{0.67}/CoMoO₄ heterostructure is more efficient to the production of H₂ and the corresponding schematic *H adsorption models are shown in Figs. S20-S21. In this light, synergistic effects between the Co₂N_{0.67} and CoMoO₄ promote HER performance through accelerating the kinetics of HER by lowering the energy barrier for water dissociation and regulating the HBE. Also, we calculated the free energy distribution of OER to explore the OER mechanism. The OER process involves multiple steps in alkaline solutions, including *OH, *O, *OOH, and O₂ intermediates (Fig. 5e, f and Fig. S22) [27,70]. Since the Mo active site in CoMoO₄ is unstable during the adsorption process, only the free energy diagrams of the Co-active site in different catalysts are shown here. As shown in Fig. 5e, for Co₂N_{0.67}/CoMoO₄, the electron transfer step from *O to *OOH (Step III) has the highest free energy gradient, indicating that it is the rate-determining step of the OER process with a theoretical overpotential of 0.17 eV. The theoretical overpotential of Co₂N_{0.67} and CoMoO₄ was 0.2 V and 0.47 V, respectively, both higher than 0.17 eV (Fig. 5f). The DFT study indicated that Co₂N_{0.67}/CoMoO₄ heterostructure composite has better OER catalytic activity. Notably, the trends of HER and OER activity in the theoretical calculations are consistent with our experimental results, which confirm that interfacial synergy is a viable mechanism for producing the high-performance electrocatalysts for overall water splitting.

3.6. Overall water splitting and Zn-H₂O cell analysis

In the light of the bifunctional HER and OER nature, we fabricated a

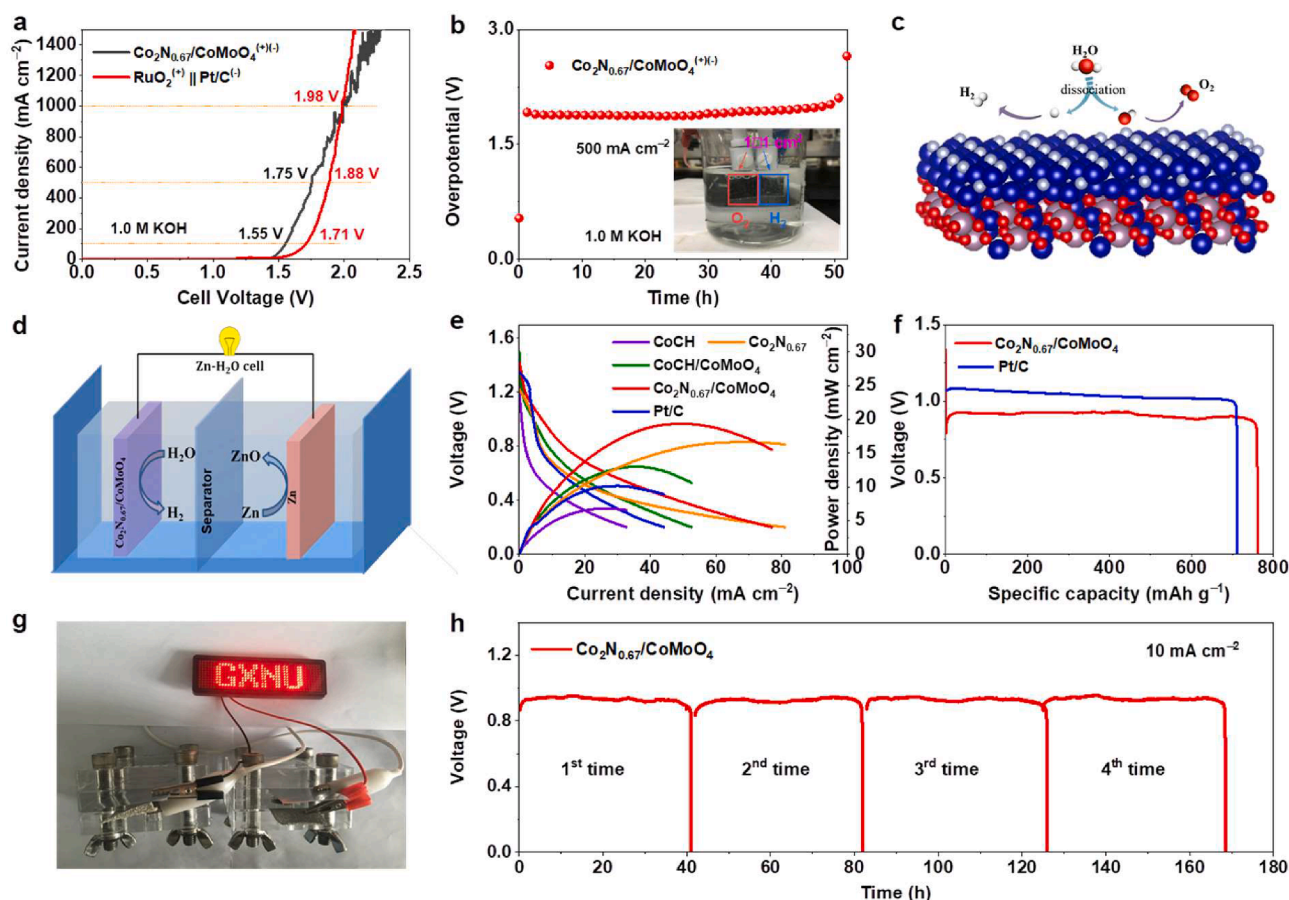


Fig. 6. (a) Polarization curves of the bifunctional Co₂N_{0.67}/CoMoO₄^(+/-) and RuO₂⁽⁺⁾||Pt/C⁽⁻⁾ catalysts for overall water splitting. (b) Chronopotentiometry curve of the bifunctional Co₂N_{0.67}/CoMoO₄^(+/-) at a current density of 500 mA cm⁻² in 1.0 M KOH. (c) Schematic illustration of the prepared Co₂N_{0.67}/CoMoO₄ for HER and OER. (d) The graphical illustration of the homemade Zn-H₂O cell. (e) Polarization curves and corresponding power density plots of CoCH, Co₂N_{0.67}, Pt/C, and other catalysts. (f) The galvanostatic discharge curve at 10 mA cm⁻² of Co₂N_{0.67}/CoMoO₄ and Pt/C. (g) A photograph of LED screen light powered by two series-connected Co₂N_{0.67}/CoMoO₄ based Zn-H₂O cells in series. (h) Long-term durability tests for Zn-H₂O cells using Co₂N_{0.67}/CoMoO₄ as the cathode.

total water splitting system using $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ as anode and cathode in a two-electrode system in 1.0 M KOH, respectively. In Fig. 6a, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4^{(+/-)}$ system requires relatively low cell voltages of 1.71, 1.88, and 1.98 V to afford the current densities of 100, 500, and 1000 mA cm^{-2} (Table S4). In particular, the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4^{(+/-)}$ outperformed $\text{RuO}_2^{(+)}||\text{Pt}/\text{C}^{(-)}$ two-electrode system at 1000 mA cm^{-2} current density. Excitingly, $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4^{(+/-)}$ displays the outstanding stability for 50 h at a high current density of 500 mA cm^{-2} (Fig. 6b). However, the stability decreased mildly after 50 h, which may be related to the structural damage of the hybrid material (Fig. S23). The excellent electrocatalytic activity is attributed to the interface promoted H_2O dissociation and the desorption of reaction intermediates on the Co-active site (Fig. 6c).

Fig. 6d shows a simple assembly diagram of the cell, in which the Zn plate as anode, $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst as cathode, and 6 M KOH and 0.2 M $\text{Zn}(\text{Ac})_2$ as an electrolyte, to further study the feasibility of $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ in Zn- H_2O cell [71,72]. In Fig. 6e, $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ has the highest peak-power density of 20 mW cm^{-2} among the other comparative catalysts. Noticeably, the cell with $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ showed a specific capacity of 750 mAh g^{-1} at the discharge current density of 10 mA cm^{-2} , which is higher than that of Pt/C based Zn- H_2O cell (700 mAh g^{-1}) (Fig. 6f). Additionally, galvanostatic discharge-charge curves were investigated at a current density of 5 mA cm^{-2} for 100 h. The $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ -based Zn- H_2O cell displayed excellent stability and reversibility (Fig. S24). More remarkably, two rechargeable Zn- H_2O cell in series can continuously light a LED screen light, also proves the potential application of the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ -based cell (Fig. 6g). The Zn- H_2O cell was deactivated after 41 h because the generated ZnO fully covered the Zn plate (anode) (Fig. 6h and Fig. S25). We then resurrected the cell by replacing a new electrolyte and zinc plate and cleaning the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ with 0.5 M H_2SO_4 . The Zn- H_2O cell showed strong discharge stability of nearly 170 h after four resurrections at a current density of 10 mA cm^{-2} . The catalyst morphology has not changed significantly (Fig. S26), demonstrating its potential for industrial applications.

4. Conclusion

In summary, a heterogeneous catalyst with interfacial structure was synthesized by hydrothermal method and ammonia etching for overall water splitting and Zn- H_2O cell. The results showed that the $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ catalyst exhibited excellent catalytic activity and high stability in both HER, OER, overall-water splitting as well as Zn- H_2O cells. The DFT calculations suggest that the self-supporting open structure of the CC in $\text{Co}_2\text{N}_{0.67}/\text{CoMoO}_4$ and the interface between $\text{Co}_2\text{N}_{0.67}$ and CoMoO_4 can enable the catalyst to expose more active sites thereby improving the intrinsic activity through adjusting the binding energy of the intermediates. As a result, the optimized catalyst exhibited outstanding catalytic performance better than most existing non-precious metal catalysts. This work provides an attractive interface engineering strategy for the rational design of non-noble metal electrocatalysts for clean energy conversion.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.cej.2022.134795>.

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